HYDROXYAPATITE AND ZINC OXIDE BASED TWO-LAYER COATING, DEPOSITED ON TI6AL4V SUBSTRATE

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At first two-layer HA-ZnO/Alg coating was obtained on the preanodized Ti6Al4V substrates by the thermal substrate deposition method (TSD). In frame of this work TSD method was at first applied for obtaining ZnO coating from aqueous alginate-containing solutions on a pre-deposited OH layer. XRD, SEM analyzes show that the biopolymer has a significant influence on the coatings formation, their morphology, texture, structure of ZnO nanoparticles. The average rate of HA deposition and ZnO deposition from alginate containing solution are 4 μ m/min and 30 μ m/min, respectively. In the presence of alginate boll-shaped ZnO microparticles are formed on the HA layer, consisting of needle-shaped crystals. Obtained coating is promising for medical implants, based on titanium and its alloys, because they potentially have both bioactivity and antibacterial activity.

KEYWORDS

Thermal substrate deposition, hydroxyapatite, zinc oxide, alginate, coating

1 INTRODUCTION

The surface modification of metal implants is one of the promising methods for improving the physico-mechanical and biological properties of these materials. The development of implants with high functionality is an urgent problem to solve. The decisive factor in the osteointegration of bioinert metal implants is the condition of their surface, which borders on the patient's bone tissue. Bioactive coatings with a developed porous surface, functionalized with biomolecules and drugs, should promote active formation of new bone tissue. Biological apatite, that is a part of the bone tissue, includes a number of the inorganic ions such as Ca²⁺, Mg²⁺, Sr²⁺, Ba²⁺, Pb²⁺, Zn²⁺, Cu²⁺, Na⁺, K⁺, Fe³⁺. Established the fact of increasing biological activity of apatite as a result of the inclusion of their composition of small number of zinc, magnesium, silver ions [Mestres 2012]. Among ceramics calcium phosphate orthophosphates, hydroxyapatite Ca10(PO4)6(OH)2 (HA), is most promising for coating of metal implants due to chemical similarity to bone mineral [Dorozhkin 2016]. Bioactive coatings based on calcium phosphates and biomolecules are applied to the surface of the metallic implants. It is known that the hybrid organic-inorganic structure initiates pliable bone. Natural polysaccharide alginate (Alg) is widely used for bone tissue engineering [Kuen 2012, Vojtko 2014], also for the stabilization of inorganic nanoparticles [Trandafilovic 2012]. The widespread use of drugs with antimicrobial action has led to the formation of microorganism resistance against antibiotics. To dissolving this problem, the inorganic substances are used for the initiation of a controlled reaction in the tissues and provision of antimicrobial activity. It is known that ZnO-based materials have a pronounced biocompatibility, are characterized by high limit strength, absolute mechanical hardness, as well as the ability to withstand the harsh operating conditions.

All available methods for the fabrication of HA coating can be divided into four groups [Yang 2009, Balara 2018, Gombar 2013, Janekova 2014, Krehel 2013, Krenicky 2012, Lesso 2010, 2014, Monkova 2013, Mrkvica 2012, Prislupcak 2014, 2016, Peterka 2014, Ragan 2012, Rimar 2014, Sebo 2012, Straka 2013, Zaborowski 2007, Zelenak 2012]: a) thermal spraying techniques, b) vapor deposition techniques, c) wet techniques and d) other deposition techniques. However, most of these methods have a weak side associated with coating surfaces of complex geometry. Besides, the modification of the metals and alloys surface before coating is one of the promising methods for improving the physical, mechanical and biological properties of the received materials [Benea 2015, Lee 2014, Suchanek 2015].

In this paper, we use a thermal substrate deposition (TSD) technique, introduced by Kuroda [Kuroda 2012] and developed by our group [Sukhodub 2003, Yanovska 2011, Yanovska 2012]. Thermal substrate method allows to obtain coatings directly from the solution at neutral pH and low temperature (40-100 °C). The method was previously used only for the calcium phosphate deposition on titanium substrates. For the first time, the method was used for the deposition of a biologically active ZnO coating on the surface of a previously deposited calcium phosphate layer.

2 MATERIALS AND METHODS

2.1 Materials

The following compounds were used for synthesis: zinc nitrate $Zn(NO_3)_2 \times 6H_2O$, anhydrous calcium chloride $CaCl_2$, orthophosphoric acid H_3PO_4 , sodium hydroxide NaOH, 25 % hydrous ammonia NH₄OH, sulfuric acid H₂SO₄, sodium fluoride NaF, sodium alginate (E401, low viscosity) manufactured by Shanghai Chemical Company Ltd, China. All components were analytically pure and used without further purification.

2.2 Methods for the obtaining of coatings

Preparation of the titanium substrate surface

Ti6Al4V template ($20\times1,8\times0.9$ mm) have been used as an experimental substrate. The sample before deposition was polished with SiC paper and washed with ethanol and deionized water in an ultrasonic cleaner, followed by anodizing for 1 hour at room temperature in an electrolyte of the next composition: $H_2SO_4 - 20\%$ w/w; NaF - 0.5% w/w to form a TiO₂ layer on the surfaces. The direct current density of 1-1.5 A/dm² was maintained for the first 5 minutes with the voltage up to 12 V. The subsequent anodizing process was carried out without adjusting the current value, its density was about 0.2 A/dm². Lead and titanium plates were used as cathode and anode, respectively. Anodized titanium substrate was annealed at 500°C for 1 hour in air in a furnace, followed with ethanol treatment and washing with deionized water.

Deposition of the be-layer HA-ZnO/Alg coating on the Ti6Al4V substrate

HA coating. Deposition of hydroxyapatite was performed on titanium substrate by the TSD method from 200 ml of the

solution, containing 0.01M CaCl₂ and 0.006M H₃PO₄. The solution pH value was adjusted to 6.52 by 10M NaOH. The anodized Ti6Al4V substrate was immersed in prepared solution and an alternating electric current (2A) was passed through the substrate for 60 minutes, resulting in a substrate heated to 95°C. To prevent spontaneous calcium phosphate formation in the whole volume when heating and to prevent a depletion of the solution on Ca²⁺ and PO₄³⁻ ions, a cooling system was used. As a result, the temperature in the whole volume of the mixture was about 25°C, while the substrate temperature was about 95°C.

$5\text{CaCl}_2 + 3\text{H}_3\text{PO}_4 + 10\text{NaOH} \rightarrow \text{Ca}_5(\text{PO}_4)_3\text{OH} + 10\text{ NaCl} + 9\text{ H}_2\text{O}$

HA-ZnO/Alg coating. The HA coating obtained above was used as the bottom layer in a HA-ZnO/Alg be-layer coating. The substrate coated with HA without detaching from the electrodes was thoroughly washed with deionized water, followed by immersion in the ZnO nanoparticles containing solution to precipitate ZnO/Alg.

The upper ZnO/Alg coating was obtained by next procedure: alginate from brown algae was used in the present study. 3 g of sodium alginate was dissolved in 100 mL of 0.01M NaOH and 4 mL of this 3% alkaline sodium alginate solution was added drop by drop to the mixture of 200 mL of 0.2M Zn(NO3)2 6H2O and 30 ml of 25% w/w ammonia under vigorous agitation to obtain the transparent alkaline solution, containing ZnO nanoparticles. The obtained solution placed into the shaker and agitated for 0.5 h (rpm 200) at room temperature.

Formation of the ZnO nanoparticles in the solution occurs according to the following chemical reaction [Zhang 2007]:

 $Zn^{2+} + 4OH^{-} \rightarrow Zn(OH)_4^{2-}$ and $Zn(OH)_4^{2-} \rightarrow ZnO + 2H_2O + 2OH^{-}$

The pre-coated by HA substrate was immersed in prepared solution and an alternating electric current (2A) was passed through the substrate for 20 minutes, resulting in a substrate heated to 80° C. This created favorable reaction conditions for the deposition of ZnO/Alg nanoparticles on the titanium substrate surface.

After the deposition, the substrate with be layer coating was gently washed with deionized water and dried at room temperature in an air atmosphere.

2.3 Investigation methods

XRD analysis

The X-ray diffraction studies of the sample crystallographic structure were performed on the automatized diffractometer DRON 3 (LTD «Burevestnik», www.bourevestnik.ru). CuK α -radiation (wavelength 0.154 nm) with θ 2 θ Bragg-Brentano geometry (2 θ is the Bragg angle) was used. The values of current and voltage on the X ray tube were 20mA and 40kV, respectively. The scan of the samples was carried out in the continuous registration mode (1/min rate) and 2 θ range from 10° to 90°. Experimental results were transferred directly to the experimental support package DifWin 1 (LTD «Etalon PTS», www.specord.ru) for the preprocessing. Identification of the crystal phases was performed using the JCPDS (Joint Committee on Powder Diffraction Standards) card catalog. More details in [Kuznetsov 2014].

SEM analysis

The surface morphology of the samples was examined using a scanning electron microscope (SEM, REMMA-102), produced by SELMI (Ukraine). Images were made in the secondary electron

mode with an accelerating voltage 20 kV and a beam current of 1–10 A.

3 RESULTS AND DISCUSSION

In this paper, the modified thermal substrate deposition method from aqueous solutions of precursors was used to obtain bioactive ZnO and HA coatings on the metal (Ti6Al4V) substrates. The application of this method for both ZnO and HA coatings is explained by the similarity of the physico-chemical processes to obtain these types of coatings. It is known that the solubility of HA in an aqueous solution decreases with increasing temperature and that the relationship between the HA solubility product, KSP/(mol×dm⁻³)⁹, and the temperature, T/K, is given by [Elliot 1994]

log KSP = -8219.41/T - 1.6657 - 0.098215×T

From our testing experiments we believe that the same properties are manifested by ZnO, so the deposition and formation of ZnO nanostructured coating on the Ti6Al4V substrate occurs approximately at the same temperatures as that for HA (90-100°C). As already noted, the deposition process involves the passage of alternating current through a metal substrate immersed in an aqueous solution. As a result, the metal substrate is heated to a temperature of about 100°C due to resistive heating (Joule heating), although the hydro process passes under atmospheric pressure.

The methods of the bioactive coatings obtaining are schematically presented in Fig.1.

One of the methods for modifying the surface of a titanium implant is the formation of oxide films (TiO_2) on it by anodizing, which improve implant anticorrosion resistance. Oxide film also is characterized by high adsorption properties. The film structure contributes to the adhesion of HA to the substrate with subsequent transformation of amorphous HA into the crystalline state [Suchanek 2017]. As mentioned above, HA coating was deposited on anodized titanium substrate.



Figure 1. Scheme for the deposition of the two-layer HA-ZnO/Alg coating on an anodized Ti6Al4V substrate using TSD method

3.1 SEM analysis.

Fig. 2 shows the SEM images of the surface morphology of titanium substrates prior and after anodizing process. The surface of the anodized sample differs from the mechanically polished one. The anodizing of titanium led to formation a compact titanium oxide film, which is confirmed by XRD analysis.



Figure 2. The surface morphology of the Ti6Al4V substrates: (A), (B) – before anodizing; (C), (D) – after anodizing

To obtain the be-layer HA-ZnO/Alg bioactive coating, a calcium phosphate (HA) layer was deposited on a titanium substrate using the TSD method. By the XRD results this layer was identified as hydroxyapatite. Fig. 3 (A, B) shows the morphology of the HA layer surface. Than ZnO/Alg layer was deposited on HA surface by the same method. Fig. 3 (C, D) shows the images of the upper layer of ZnO/Alg deposited on the HA. ZnO particles are ball-shaped, with a complex internal structure, represented in the insert (Fig. 3, D).



Figure 3. Morphology of the obtained by TSD method coatings: (A), (B) - HA coating at different magnifications; (C), (D) - two-layer HA-ZnO/Alg coating: upper ZnO layer on HA at different magnification

3.2 XRD analysis

The X Ray diffraction patterns of the samples are present on Fig. 4. The major peaks of ZnO, and HA on the corresponding diffractograms are marked with Miller indexes. The Ti6Al4V anodized substrate diffractogram shows peaks belonging to TiO₂ (JCPDS Card No 76-0322 Rutile).



Figure 4. Diffraction patterns from samples. HA coating: (0 0 2) and (2 1 1) lines are the main peaks of HA, marked with \blacklozenge , the TiO2 substrate peaks are marked with \blacklozenge ; two-layer HA-ZnO/Alg coating with ZnO/Alg as upper layer: ZnO peaks are marked with Miller indexes

The phase composition of the HA coated substrate (Fig.4) consists of a HA (JCPDS 9-432). The broad peak for HA indicate that coating is nanocrystalline. The presence of Ti peaks (JCPDS 44-1294) on a diffractogram indicates a low thickness of the HA coating. The thickness of the HA deposited for 60 minutes is 250 μ m. The rate of calcium phosphate deposition of 4 μ m/min is significantly lower, than zinc oxide in the case of ZnO/Alg coating. HA is calcium deficient, its atomic ratio Ca/P = 1.64 by RFA analysis.

The diffractogram for upper ZnO/Alg layer of HA-ZnO/Alg coating shows intensive peaks of zinc oxide, which overlap the HA ones. Especially intensive peak (002) (up to 5200a.u.) suggests the predominant growth of ZnO nanoparticles along c-axis. The XRD peaks for ZnO/Alg coating, are indexed to hexagonal ZnO phase and in accordance with the JCPDS Card No 36-1451.

It is obvious that the texture is present for upper layer -ZnO/Alg coating. The Harris method (Harris texture index, HTI) was used for its quantitative estimation. The crystallite sizes by Scherrer (L) were evaluated for all major peaks of ZnO phase, as well as its unit cell parameters (*a* and *c*, nm) using (1 0 3) and (0 0 2) peaks respectively. The good resolution of (0 0 2) and (0 0 4) peak allowed to separate the contributions from small sizes of coherent scattering regions (L_{WH}) and the presence of microstrains (ϵ) into the peaks broadening. The separation of these contributions was performed with the Williamson–Hall method [Klug 1974]. These results are present in Table 1.

 Table 1.
 ZnO crystal structure parameters for the two-layer HA-ZnO/Alg coating on the Ti6Al4V substrate, obtained by TSD method

Sample	Value	(100)	(002)	(101)	(110)	L _{WH} . nm ε. 10 ⁻³	a. nm c. nm
HA- ZnO/Alg	L. nm	13.1	38.8	8.9	9.6	110.9	0.3217
	HTI	0.040	2.658	0.061	0.070	2.921	0.5188

The crystal structure of the be-layer HA-ZnO/Alg coating corresponds to other samples, but shows the highest crystallite sizes and amount of microstrains in (0 0 l) direction. This fact reaffirms the influence of the surface on the structure of ZnO coating.

In [Kuroda 2012] study, an *in vivo* investigation of HA/collagen composite coatings on titanium substrates, obtained by TSD

method, was carried out. It has been found that such composite coatings exhibit high osteoconductivity in the cortical bone. This tendency also occurs for the cancellous bone part. Moreover, HA is known as bioactive, osteoconductive material. Since the HA coating obtained in our work was deposited in similar conditions to the conditions described above, we believe that they can also potentially positively affect the osteoconductivity in the formation of bone tissue.

The study examined the surface of the intraosseous part of screw dental implants in the following areas (SS): the top of the thread, the ramp of the thread, the base of the thread, the distance between the threads of the thread, the groove (Fig. 5). To study the morphology of the coatings, their phase and elemental composition, scanning electron microscopy, X-ray spectral analysis and X-ray structural analysis were used.



Figure 5. Helical implant with uniform thread and apical groove: a) Medical implant; b) model for research



Figure 6. x-ray photograph of samples (located SS)

Element	Weight%	Atomic%	
СК	25.89	37.73	
ОК	42.93	46.97	
Na K	0.76	0.58	
Al K	0.21	0.14	
Si K	0.55	0.34	
РК	9.94	5.62	
CI K	0.37	0.18	
КК	0.22	0.10	
Ca K	19.07	8.33	
Ti K	0.07	0.03	
Totals	100.00		

Table 2	Composition	of conting	under EDC
Table 2.	Composition	of coating	under EDS

X-ray structural phase analysis of the samples allowed us to obtain diffraction patterns with characteristic peaks of certain phase components (Fig. 6). In tab. 2 shows the results of decoding x-ray photograph of samples.

Studies performed by scanning electron microscopy showed that a surface with a high level of roughness is formed, containing a large number of microcracks (Fig. 7).







Figure 7. Hydroxyapatite coating is the thinnest polymorphic surface structure formed during the solidification of hydroxyapatite powder particles

The surface of the coating is homogeneous, dense, contains visible defects, cracks and chips. The applied coating repeats the surface relief of the substrate. There is a decrease in the areas of Ca₃(PO₄)₂ (the product of decomposition of HA). At the same time, calcium pyrophosphate γ -Ca₂P₂O₇, CaO, various calcium bicarbonates and triple calcium hydrogen phosphate Ca₃(PO₄)₂H₂O are fixed. The crystallinity of the coating is mainly determined by the presence of the hydroxylapatite phase,

calcium phosphate and ternary calcium phosphate, the latter having the largest crystallite sizes, on the borders of which nanoscale calcium phosphate particles are located (Fig. 7).

4 CONCLUSIONS

At first two-layer HA-ZnO/Alg coating was obtained on the preanodized Ti6Al4V substrates by the TSD method. It has been experimentally proved that this method can be effectively applied to obtain not only HA coating, but ZnO/Alg coating on a surface of HA inorganic layer. Natural biopolymer alginate has a significant influence on the coatings formation, their morphology, texture, structure of ZnO nanoparticles. The average rate of the ZnO deposition from alginate containing solution is 30 μ m/min, the thickness of the ZnO / Alg coating, is 600 μm. The two-layer HA-ZnO/Alg coating is potentially bioactive due to HA and exhibits antimicrobial activity through the zinc oxide nanoparticles. For this coating an increase in the ZnO crystallites size and microstrains along the (0 0 l) direction are characteristic. The obtained coatings are promising for medical implants, based on titanium and its alloys, because they potentially have both bioactivity and antibacterial activity.

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