SUPERHYDROPHOBIC STAINLESS STEEL SURFACE BY TWO-STEP NS LASER PROCESSING

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Growing demand for superhydrophobic surfaces in recent years is associated to many attractive science and engineering applications including self-cleaning, anti-icing and anti-corrosive behaviours. Stainless steel type AISI 316L is one of the most versatile and widely used engineering material in industries. Inspired by the "lotus effect" nano/microstructures has been fabricated by direct laser writing method with nanosecond laser source using two ablation regimes. Primarily, microstructures were fabricated with a tightly focused beam and covered by nano-scale structures by defocused laser beam in the second fabrication step. However, freshly prepared laser patterned metal surface shows hydrophilic behaviour. The hydrophilic to superhydrophobic transformation takes several days or weeks by aging technique in atmospheric condition. In this study, the transition time has been drastically reduced by high vacuum processing technique. Wetting properties with respect to laser processing parameters and surface morphology were examined and found to be consistent for large droplet volumes.

KEYWORDS

Laser micromachining, superhydrophobicity, functional surfaces, stainless steel.

1 INTRODUCTION

Fabrication of superhydrophobic surfaces draw a lot of attention in recent years due to many potential applications in science and industry including production of anticorrosion [Khorsand et al. 2016], anti-bacteria [Zhao et al. 2018], anti-icing [Bharathidasan et al. 2014] or self-cleaning surfaces [Bhushan et al. 2009].

Inspired by the nature, many fabrication methods have been developed to replicate extremely water repellent lotus leave surface covered with hierarchical dual-scale micro and nanostructures [Liu et al. 2010] including lithography [Shiu et al. 2004], chemical etching [Cremaldi et al. 2018], chemical vapor deposition [Kamal et al. 2015], electrodeposition [Liu et al. 2016], sol-gel [Lakshmi et al. 2009], thermal embossing [Toosi et al. 2016], and laser surface texturing [Jagdheesh et al. 2019].

Among these techniques, laser surface texturing offers flexible, non-contact method for fast and efficient nano and microstructure fabrication over large areas without need of additional chemical treatment. Moreover, the fabrication speed can be scaled up several times using polygonal scanners [De Loor 2013], multi-beam and interference processing [Aguilar-Morales et al. 2018] reaching 0.9 m²/min [Lang et al. 2016].

However, metal oxide layer formed after laser processing on the top of the geometries results in a hydrophilic nature of processed sample. Exposed to atmospheric conditions, several days to a few weeks are required for transition from hydrophilic to hydrophobic or superhydrophobic state [Kietzig et al. 2009]. This time can be dramatically reduced to a several hours by vacuum processing introduced by Jagdheesh [Jagdheesh et al. 2017].

Wettability of a rough surface can be described by two wellknow theories. According to Wenzel model [Wenzel 1936] for a fully-wetted rough surface, an originally hydrophilic surface (contact angle (CA) < 90°) becomes more hydrophilic with increased roughness and hydrophobic surface (CA > 90°) properties are also enhanced with increased roughness. Contrarily, Cassie-Baxter model [Cassie et al. 1944] describing the surface able to keep air pockets inside surface features shows improvement of hydrophobic character with increasing roughness even for originally hydrophilic flat samples.

In this paper, a nanosecond laser source is used to produce a combination of micro and nanostructures in a two-step fabrication process on AISI 316L steel, which is commonly used in industrial contexts such as medical implants, precision mechanics, marine applications and food handling. In the first step microstructures are made by a tightly focused beam and covered by nanoscale-protrusions in the second step using defocused laser beam. Wettability of textured surfaces is analyzed with respect to geometry and processing parameters.

2 MATERIAL AND METHODS

AlSI 316L stainless steel plates were used as received (Ra \sim 0.3 µm) and treated in air with nanosecond Ytterbium fiber laser with a wavelength of 1062 nm and output power of 12 W, pulse duration of 3 ns and adjustable repetition rate in a rage of 100-1000 kHz. Galvanometric scanner system with 163 mm F-theta lens was used for beam displacement over the sample with the maximum available scanning speed of 700 mm/s.

Sample was processed in two steps in order to fabricate hierarchical dual-scale structures. In the first step, a tightly focused beam with 39 μ m in diameter was used to produce micropillars by moving the laser beam in a rectangular grid path with a hatch distance (HD) in a rage of 30 – 50 μ m and with up to 400 overscans. Laser repetition rate and pulse energy were optimized during the process for best quality micropillars. In the second step, beam was defocused by moving the sample closer to the laser source by a defocusing distance (*d*) and scanned line by line over the whole sample with a line hatch distance of 0,5 μ m and single overscan, which results in a formation of nanoscale features on the top of micropillars as it was demonstrated in our previous work [Hauschwitz et al. 2019].

Samples were put into vacuum chamber immediately after laser treatment and kept in high vacuum ($2.6 \cdot 10^{-6}$ Pa) for 12 hours to promote favourable chemical changes to decrease transition time between hydrophilic and hydrophobic state.

Surface morphology were examined with a laser confocal microscope Olympus OLS5000. Wettability of laser-treated samples were measured with optical contact angle measuring device OCA 15EC, Data Physics Instruments using 10 μ l droplets of deionized water. The results were acquired through the average of 5 measurements on different locations for every sample surface.

3 RESULTS AND DISCUSSION

Thermal processes including rapid heating and boiling of the melt are connected with nanosecond ablation [Leitz et al. 2011], resulting in the melt ejection and forming of a recast layer around the interaction zone. Therefore, laser and processing parameters have to be properly optimized to ablate well-defined structures with minimal unwanted thermal effects.

Tightly focused beam with a train of nanosecond pulses is used in the first fabrication step with the aim to produce high quality and melt-free micropillars with a maximum possible depth. The quality of produced micropillars was optimized by changing repetition rate, pulse energy and hatch distance, while keeping the power and scanning speed at the maximum level, as depicted in the Fig. 1. The micropillar height evolution in a dependence on repetition rate and pulse energy is shown in the Fig. 2.



Figure 1. Representative confocal images of fabricated micropillars with the repetition rate in a rage of 100 kHz to 1 MHz and HD in a rage of 30 μ m to 50 μ m.



Figure 2. Micropillar height dependence on the repetition rate and pulse energy for 30 $\mu m,$ 40 μm and 50 μm hatch distance.

As can be observed in Fig. 1, too small hatch distance, below 40 μ m can result in an irradiation of the whole area leading to a smaller pillar high, especially when a high number of pulses is hitting the same spot due to high repetition rate > 800 kHz, as demonstrated in the Fig. 2. This behavior can be explained by the accumulation effect between consecutive laser pulses. Due to gaussian distribution of the laser beam, the edges of the beam are below ablation threshold. However, when high number of pulses are hitting the same spot due to high repetition rate, accumulation between consecutive laser pulses pulses leads to temperature increase on the target between these pulses and results in

a decrease in ablation threshold [Weber et al. 2014]. Therefore, a bigger part of a gaussian beam is above the threshold and material can be ablated in a wider area. As a result, width and depth of produced structures is reduced for high repetition rates, as depicted in Fig. 1. Contrarily, due to same pump power, pulse energy is decreasing with higher repetition rate decreasing the effective beam diameter. However, the accumulation effect is much more significant, as demonstrated in Fig. 1.

Therefore, the highest pillars were demonstrated for the lowest used repetition rate of 100 kHz with enough spacing between pillars (HD of 50 μ m), as shown in Fig. 2. According to the analysis of suitable microstructure geometry for superhydrophobicity [Kong et al. 2019], the micropillar height is an important aspect which should generally improve water repellency when increased. Hence, surface with the highest micropillars (Figure 3) was selected as the optimal surface for the further fabrication step.



Figure 3. Confocal image of the optimal surface topography after the first fabrication step with 100 kHz repetition rate and 50 μm hatch distance, 400 overscans.

In the second step, defocused laser beam is introduced on the surface with previously fabricated micropillars in order to produce nanoscale features on the top of them by gently melting the surface without removing the structures. For this purpose, hatch distance was decreased to 0.5 μ m to introduce high enough line to line overlap for melting the surface with defocused beam. Repetition rate and defocusing distance had to be further optimized to prevent overwriting microstructures from the first fabrication step, as depicted in Fig. 4. Power and scanning speed were kept at maximum for higher processing rates.





Figure 4. Representative confocal images of structure evolution for repetition rate in a range of 300 - 1000 kHz and defocusing distance in a range of 0.2 - 1 mm, during second fabrication step.

Higher pulse energy (RR < 500 kHz) and short defocusing distance (d < 0.6 mm) results in high intensity on a surface and removal of micropillars from the surface, as can be observed in the Fig. 4. By decreasing repetition rate or increasing defocusing distance above these values, gently melted micropillar surface, covered with nanoscale features can be fabricated. Contrarily, too small intensity (RR > 1 MHz, d > 1 mm) only heats up the surface without melting and no nanostructures are formed (results not shown). Optimal surface after second fabrication step is depicted in Figure 5.





All samples are superhydrophilic after processing due to formation of metaloxydes with high surface energy. It takes a few days to few weeks for samples kept in an atmospheric conditions to reach superhydrophobic properties [Kietzig, Hatzikiriakos and Englezos 2009]. Therefore, immediately after laser processing samples were put in a vacuum chamber and kept in the high vacuum conditions (2.6-10-6 Pa) for 12 hours. This vacuum processing technique introduced by Jagdheesh [Jagdheesh, Diaz, Marimuthu and Ocana 2017] significantly reduces the time needed for wettability transition from hydrophilic to superhydrophobic state by preventing water molecules to form a passivation layer which slows down the evolution of non-polar hydrocarbon layer primarily responsible for superhydrophobic behaviour.

After vacuum processing, wettability of selected best quality hierarchical structures was analysed with 10 μl

droplets of deionized water and sessile drop technique. These results are depicted in the Fig. 6.



Figure 6. Contact angle evolution with the defocusing distance for different repetition rates.

Contact angles were measured only for defocusing distances above 0.4 mm. Below this value, most of the micropillars were removed during the second fabrication step. As depicted in Fig. 6, there is an optimal defocusing distance around 0.6 mm when contact angle reaches its maximum for all used repetition rates. A sharp decrease in contact angle is observed for defocusing distance above 0.8 mm. It can be explained by the drop in intensity on the surface which is not high enough for creation of nano-scale features. The highest contact angle of 168° was reached for the surface fabricated with 1 MHz and 0.6 mm defocusing distance. Droplet deposited on this surface can be observed in Fig. 7. The detail look on the contact area reveals that droplet is deposited only on the top of produced micropillars and it is not wetting the whole surface. Therefore, surface is in a Cassie-Baxter state.



Figure 7. Detail of a droplet deposited on an extremely water repellent surface fabricated with 1 MHz and 0.6 mm defocusing distance in a second fabrication step.

4 CONCLUSION

Well-defined micro pillars covered with nanoscale protrusions were fabricated with properly optimized nanosecond laser source using the two-step fabrication technique. The height and width of micropillars can be controlled by repetition rate and hatch distance during the first fabrication step. In the second step, nanoscale protrusions are fabricated on the top of micropillars without removing the original structure, when defocusing distance and repetition rate are properly optimized. The wettability transition time from freshly processed hydrophilic sample to superhydrophobic was significantly reduced by vacuum processing technique. The best wettability result with contact angle of 168° was reached for surface produced with 100 kHz and 50 μ m hatch distance in the first step, followed by the second processing step with 1 MHz and 0.5 μ m hatch distance, defocused for 0.6 mm above the surface.

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